Synthesis and Characterization of Novel Arylene Cyanate Resins

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1. Introduction

Wide bandgap materials such as silicon carbide are quite attracting for power semiconductor modules. Since such modules require quite high operating temperature at around 250 °C, the molding materials also require extremely high thermal stability as well as good processability. Conventional epoxy resins typically survive only below 150 °C; therefore, the development of novel thermosetting resins is required.

As cyanate ester (CE) resin is a thermosetting resin that has high thermal stability, it could be a promising material for the molding material of power semiconductors [1-3]. One major problem of CE resin is, however, low processability at the monomer stage. The processing temperature must be largely lower than the curing temperature considering the fabrication procedure of power semiconductor modules. In this context, it is quite important to develop the CE monomer that has low deformation temperature before curing and good thermal stability after curing.

In the current study, we investigate the mixtures of two different aromatic CE monomers. We expect that the deformation temperature of the monomer mixture becomes lower than that of pure monomer if a eutectic mixture could be prepared [4]. The effect of supercooling is also of interest since aromatic bulky structure might affect the conformational dynamics [5]. Fig. 1 shows the structure of the CE monomers designed in the current study (CE124, CE12 and CE13). We selected aromatic main chain considering the thermal stability after curing and designed three different but similar structures considering the mutual solubility. Corresponding model compounds (M124, M12 and M13) with methoxy terminal groups were also studied to clarify the deformation and supercooling properties of the mixtures under the condition that is not affected by the curing reaction.

2. Experimental

Tetrahydrofuran (THF) was used after distillation from sodium. Triethylamine (TEA), N,N-dimethylformamide (DMF) and the other chemicals were used as received.

M124: 1H NMR (DMSO-d6, 400 MHz, δ, ppm) 8.11 (d, 2H, Ar-H), 7.92 (dd, 4H, Ar-H), 7.55 (d, 1H, Ar-H), 7.49 (ss, 1H, Ar-H), 7.32 (dd, 1H, Ar-H), 7.14 (d, 2H, Ar-H), 6.99(dd, 4H, Ar-H), 3.87 (s, 3H, -OCH3), 3.80(s, 6H, -OCH3); IR (KBr, cm⁻¹) 3076, 3005, 2933, 2821, 1744, 1733, 1581, 1513, 1464, 1440, 1320, 1250, 1188, 1142, 1105, 1063, 1024, 1007, 847, 691, 612, 585.

M12: 1H NMR (DMSO-d6, 400 MHz, δ, ppm) 7.91 (d, 4H, Ar-H), 7.48-7.39 (m, 6H, Ar-H), 7.00 (d, 4H, Ar-H), 6.99(dd, 4H, Ar-H), 3.87 (s, 3H, -OCH3), 3.80(s, 6H, -OCH3); IR (KBr, cm⁻¹) 3076, 3005, 2933, 2821, 1744, 1733, 1581, 1513, 1464, 1440, 1320, 1250, 1188, 1142, 1105, 1063, 1024, 1007, 847, 691, 612, 585.

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3010, 2979, 1740, 1727, 1605, 1512, 1495, 1469, 1243, 1171, 1100, 1067, 1021, 851, 760, 692.

M13: $^1$H NMR (DMSO-$d_6$, 400 MHz, $\delta$, ppm) 8.10 (d, 4H, Ar-H), 7.55 (t, 1H, Ar-H), 7.28 (sss, 1H, Ar-H), 7.23 (dd, 2H, Ar-H), 7.14 (d, 4H, Ar-H), 3.88 (s, 6H, -OCH$_3$); IR (KBr, cm$^{-1}$) 3062, 2836, 1725, 1604, 1510, 1478, 1455, 1321, 1251, 1165, 1130, 1066, 1026, 847, 763.

CE124, CE12 and CE13 were synthesized by the routes shown in Scheme 2.

CE124: $^1$H NMR (DMSO-$d_6$, 400 MHz, $\delta$, ppm) 8.33 (d, 2H, Ar-H), 8.14 (dd, 4H, Ar-H), 7.69 (d, 2H, Ar-H), 7.65 (d, 1H, Ar-H), 7.56 (dd, 5H, Ar-H), 7.45 (dd, 1H, Ar-H); IR (KBr, cm$^{-1}$) 3107, 3068, 2286, 2258, 2239, 1754, 17401597, 1502, 1416, 1256, 1252, 1186, 1162, 1144, 1107, 1070, 1012, 851, 754.

CE12: $^1$H NMR (DMSO-$d_6$, 400 MHz, $\delta$, ppm) 8.14 (d, 4H, Ar-H), 7.56 (d, 4H, Ar-H), 7.57-7.45 (m, 4H, Ar-H); IR (KBr, cm$^{-1}$) 3111, 3072, 2287, 2260, 1742, 1599, 1500, 1458, 1418, 1305, 1275, 1260, 1205, 1175, 1158, 1105, 1070, 1014, 845, 752.

CE13: $^1$H NMR (DMSO-$d_6$, 400 MHz, $\delta$, ppm) 8.30 (d, 4H, Ar-H), 7.67 (d, 4H, Ar-H), 7.58 (t, 1H, Ar-H), 7.39 (sss, 1H, Ar-H), 7.30 (dd, 2H, Ar-H); IR (KBr, cm$^{-1}$) 3107, 3075, 2292, 2263, 1733, 1733, 1595, 1499, 1479, 1417, 1308, 1273, 1212, 1164, 1143, 1073, 1011, 891, 851, 755.

The deformation temperatures of the model compounds and CE monomers were determined by thermomechanical analysis (TMA) using SII TMA/SS6000 with a quartz rod probe at 100 mN.

Differential scanning calorimetry (DSC) was carried out using SII EXSTAR DSC6000 under N$_2$ flow.

Polarized optical microscopy (POM) was carried out using Olymnap BX51 equipped with a heating stage (Mettler Toledo FP90 and FP82HT).

Thermogravimetric analysis (TGA) was carried out using SII TG/DTA7300 under N$_2$ flow.

FT-IR spectra were obtained with a JASCO 4100 spectrometer by the KBr pellet method.

3. Results and discussion

The model compounds were mixed by dissolving them into dichloromethane and evaporating the solvent. Fig. 2 shows the photo images of the mixtures after the evaporation. Most of samples were collected as white powder, but M124(40)/M12(60), M124(50)/M13(50) and M124(40)/M13(60) kept liquid form. These liquid samples turned into white solid powder within 2 weeks at room temperature; therefore, it is likely that the liquid form appeared by supercooling. The crystallization of the compound was probably inhibited by mixing two different compounds, or the bulky aromatic structure inhibited the conformational relaxation. This may be an instructive phenomenon considering the design of processable monomers although the life of supercooling with the current sample is too short to apply this property to processable cyanate monomers.

The deformation temperatures of thus obtained powder mixtures were evaluated by TMA. Fig 3 shows the deformation temperatures of the mixed model compounds and pure model compounds. The deformation temperature of pure M124, M12 and M13 are 143, 126 and 134°C, respectively. The mixed samples shows lower deformation temperatures. No matter whether the sample shows
supercooling property, the mixed samples show similar deformation temperatures at around 103 and 108 °C for the M124/M12 and M124/M13 mixtures, respectively. This tendency is probably because the deformation temperature of the mixed sample reflected the melting point of a eutectic mixture. It is concluded that mixing two compounds would contribute to lower deformation temperature and this information would be useful to design a CE monomer with a relatively low processing temperature.

The CE monomers were mixed by dissolving them into dichloromethane and evaporating the solvent. None of the tested samples showed supercooling property in contrast to the model compounds. The deformation temperatures of thus obtained powder mixtures were evaluated by TMA. The deformation temperature of pure CE124, CE12 and CE13 are 179, 120 and 145 °C, respectively. The mixed samples show lower deformation temperatures. They show similar deformation temperatures at around 102 and 136 °C for the M124/M12 and M124/M13 mixtures, respectively. This tendency is probably because the deformation temperature of the mixed sample reflected the melting point of a eutectic mixture. It should be noted that the deformation temperature of the CE124/CE12 mixtures is considerably lower than those of pure monomers.

The deformation and curing behavior of the mixed CE monomers was characterized by DSC and POM. The CE monomer mixtures were cured by the following manner. A sample was placed on an aluminum pan and heated up to 200 °C under evacuation. And then N2 gas was introduced and the temperature was kept at 280 °C for 4 h. The CE monomer mixtures were cured by the following manner. A sample was placed on an aluminum pan and heated up to 200 °C under evacuation. And then N2 gas was introduced and the temperature was kept at 280 °C for 4 h. Fig. 7 shows the FT-IR spectra of the CE124(40)/CE12(60) mixture before and after the curing procedure. The peaks at around 2240 cm⁻¹ due to O-C≡N groups significantly decreased and new peaks due to triazine rings appeared at 1373...
and 1559 cm\(^{-1}\) after the curing.

The thermal stability of the cured resins was characterized by TGA. Table 1 summarizes the weight loss at 350 °C and the temperatures at which 5% weight loss was recorded (\(T_{5d}\)). In all tested samples, the weight loss at 350 °C is below 2 wt% and \(T_{5d}\) is at around 400 °C. It could be concluded that the CE resins studied in this work have quite promising thermal stability considering the application for the molding materials.

Table 1 Thermal stability characterized by TGA

<table>
<thead>
<tr>
<th>Sample</th>
<th>Weight loss at 350 °C (%)</th>
<th>(T_{5d}) (°C)</th>
</tr>
</thead>
<tbody>
<tr>
<td>CE124</td>
<td>0.41</td>
<td>418.5</td>
</tr>
<tr>
<td>CE12</td>
<td>2.86</td>
<td>375.8</td>
</tr>
<tr>
<td>CE13</td>
<td>1.23</td>
<td>405.0</td>
</tr>
<tr>
<td>CE124/CE12</td>
<td></td>
<td></td>
</tr>
<tr>
<td>80/20</td>
<td>1.50</td>
<td>405.0</td>
</tr>
<tr>
<td>60/40</td>
<td>1.60</td>
<td>399.3</td>
</tr>
<tr>
<td>50/50</td>
<td>1.79</td>
<td>394.3</td>
</tr>
<tr>
<td>40/60</td>
<td>1.46</td>
<td>395.8</td>
</tr>
<tr>
<td>20/80</td>
<td>1.41</td>
<td>390.4</td>
</tr>
<tr>
<td>CE124/CE12</td>
<td></td>
<td></td>
</tr>
<tr>
<td>80/20</td>
<td>1.57</td>
<td>410.8</td>
</tr>
<tr>
<td>60/40</td>
<td>1.66</td>
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<td>411.2</td>
</tr>
<tr>
<td>20/80</td>
<td>0.97</td>
<td>409.2</td>
</tr>
</tbody>
</table>

**Conclusion**

We have designed and synthesized novel arylene CE monomers and clarified the effect of mixing two different CE monomers. The deformation temperature was successfully lowered by mixing two monomers. The most promising combination in the current samples is the CE124/CE12 mixture, which shows the deformation temperature at 102 °C and the curing temperature at around 170-280 °C. All tested samples in this study show promising thermal stability. Further study will be done to achieve even lower deformation temperature by optimizing the structure and composition of the CE monomers.

**References**